DE LA RECHERCHE À L'INDUSTRIE

(Paris, France) **Saclay**

Magneto-Crystalline anisotropy of Fe, Co and Ni slabs : A benchmark from DFT and Tight-Binding models

Ludovic Le Laurent¹ , Cyrille Barreteau**¹** , Troels Markussen**² (ludovic.le-laurent@cea.fr) ¹CEA Saclay, ²Synopsys**

www.cea.fr

MAGNETIC ANISOTROPY

In magnetic materials :

- \cdot Easy-axis magnetization : $E_{syst} = f(\vec{m})$
- Switchable domains \rightarrow Data storage.

- → **Nanoscale systems**.
- → **Smaller ? Thermal fluctuations…**
- \rightarrow Find system with large Anisotropy.

Why an easy-axis instead of another ?

 \rightarrow We study the difference of total energy under two different magnetizations :

 $MAE = E_{tot}(m_1) - E_{tot}(m_2)$

- Origin is twofold :
- Shape anisotropy.
- Magneto-crystalline anisotropy (MCA).

TWO ANISOTROPIES

- Shape anisotropy **CONCA**
- Two magnetizations \vec{m}_i, \vec{m}_j , as a magnetic dipole.

$$
E_{\text{dip}} = \frac{\mu_0}{8\pi} \sum_{i \neq j} \frac{1}{r_{ij}^3} \left[\overrightarrow{m_i} \cdot \overrightarrow{m_j} - 3 \frac{(\overrightarrow{r_{ij}} \cdot \overrightarrow{m_i})(\overrightarrow{r_{ij}} \cdot \overrightarrow{m_j})}{r_{ij}^2} \right]
$$

Dipole-dipole interaction.

$$
E_{\text{dip}} = \frac{\mu_0}{8\pi} \sum_{i \neq j} \frac{m_i m_j}{r_{ij}^3} (1 - 3\cos^2(\theta_{ij}))
$$

- \rightarrow **in-plane** magnetization.
- Independent of crystalline structure.
- Depends only on quantity of matter and shape.
- At big scales, wins always !
-
- H = Schrödinger + Zeeman + Mass velocity + Spin-orbit \rightarrow relativistic corrections.
- Quantum effects :

$$
H^{SOC} = \sum_{i} \xi_{i,d} \vec{L}_i \cdot \vec{S}_i
$$

- SOC breaks spherical invariance.
- MCA depends on symmetry, nature of atoms.
- Dominating at surfaces and interfaces.

TWO ANISOTROPIES

- Shape anisotropy **MCA**
- Two magnetizations \vec{m}_i, \vec{m}_j , as a magnetic dipole.

$$
E_{\text{dip}} = \frac{\mu_0}{8\pi} \sum_{i \neq j} \frac{1}{r_{ij}^3} \left[\overrightarrow{m_i} \cdot \overrightarrow{m_j} - 3 \frac{(\overrightarrow{r_{ij}} \cdot \overrightarrow{m_i})(\overrightarrow{r_{ij}} \cdot \overrightarrow{m_j})}{r_{ij}^2} \right]
$$

Dipole-dipole interaction.

$$
E_{\text{dip}} = \frac{\mu_0}{8\pi} \sum_{i \neq j} \frac{m_i m_j}{r_{ij}^3} (1 - 3\cos^2(\theta_{ij}))
$$

- \rightarrow **in-plane** magnetization.
- Independent of crystalline structure.
- Depends only on quantity of matter and shape.
- At big scales, wins always !

- H = Schrödinger + Zeeman + Mass velocity + Spin-orbit \rightarrow relativistic corrections.
- Quantum effects :

$$
H^{SOC} = \sum_{i} \xi_{i,d} \vec{L}_i \cdot \vec{S}_i
$$

- SOC breaks spherical invariance.
- MCA depends on symmetry, nature of atoms.
- Dominating at surfaces and interfaces.

HOW TO CALCULATE MCA (1/2)

 Brute force method (self-consistent) : *MCA*=*Einplane*−*Eoutofplane* where the two energies are obtained from SCF calculation including SOC.

In principle « exact » but very time consuming and hard to converge One should use penalization techniques to obtain energy for any direction.

 Force Theorem method : The variation of energy between a SCF calculation without SOC and with SOC is just the band energy variation.

$$
MCA^{FT} = \int_{0}^{E_F^1} E \rho_1(E) dE - \int_{0}^{E_F^2} E \rho_2(E) dE
$$

$$
\Delta E_{\text{tot}} = \Delta E_{\text{band}}
$$

 $MCA^{F T g c} = \int (E - E_F) \Delta \rho(E) dE$, at fixed chemical potential. *EF* $(E-E_F)\Delta\,\rho(E)dE$

Very « fast » and numerically stable but cannot be applied to systems with too large SOC.

HOW TO CALCULATE MCA (2/2)

- Three computational tools : home-made Tight-Binding (TB), Quantum Espresso (DFT QE) and Quantum ATK (DFT QATK).
	- \rightarrow experimentally, MCA really weak, numerically too ! We have to compare different approaches.

Semi-empirical Tight-Binding fitted on Density Functionnal Theory data :

- On-site
- Hopping
- Overlap
- Stoner parameter
- SOC

 \rightarrow Really fast computational time.

Quantum Espresso : *Ab initio* expanded on plane waves.

 \rightarrow complete basis describing the whole system plus void, but huge computational cost.

TB DFT

Quantum ATK : *Ab initio* expanded on Local Atomic Orbitals.

 \rightarrow localized basis with a really good description of partial system, fast computational time.

TOTAL MCA FOR SLABS : IRON, COBALT AND NICKEL (1/2)

Mesh SCF : 25*25 k-points / Mesh NSCF : 50*50 k-points

TOTAL MCA FOR SLABS : IRON, NICKEL AND COBALT (2/2)

Fe bcc vs thickness (QATK)

Ni fcc vs thickness (QATK)

Co fcc and hcp vs thickness (QATK)

MCA BY LAYERS FOR SLABS : IRON, NICKEL AND COBALT(1/2)

• How to explain oscillations ? It can be interesting to decompose MCA, since we can write :

$$
MCA^{FIGc} = \int_{0}^{E_{F}} (E - E_{F}) \Delta \rho(E) dE = \sum_{i, \lambda, k} \int_{0}^{E_{F}} (E - E_{F}) \Delta \rho_{i, \lambda, k}(E) dE
$$

- \rightarrow decomposition by layers of MCA, index i
- \rightarrow decomposition by orbitals, index λ
- \rightarrow large thickness behavior
- \rightarrow MCA in Brillouin zone, index k
- \rightarrow band structure

Co hcp vs site (TB)

MCA BY LAYERS FOR SLABS : IRON, NICKEL AND COBALT(2/2)

| PAGE 10

MCA ANALYSIS IN TIGHT-BINDING : LARGE THICKNESS BEHAVIOR FOR COBALT

MCA ANALYSIS IN TIGHT-BINDING : K-SPACE FOR BULK COBALT

MCA ANALYSIS IN TIGHT-BINDING : BAND STRUCTURE FOR BULK COBALT

 \cdot Most important contributions : high symmetry points H and Γ !

 \rightarrow tangential bands at Fermi level in one orientation but not in the other one.

• According to the level, MCA varies rapidly !

BAND STRUCTURE IN DFT FOR BULK COBALT

Differences observed in MCA between codes = Slightly differences in band structures according to Fermi level !

FEW IDEAS ABOUT TUNING MCA

Modifying the filling is a way to pilot MCA from in-plane to out-of-plane orientation :

- Electric field $\rightarrow 1.10^{\prime}10$ V/m for only 0,05 electrons...
- Molecules and charge transfer.

AKNOWLEDGEMENTS

FET open Grant No. 766726

COncepts and tool**S** in **M**olecular spintron**ICS**

This project has received funding from the European Union's Horizon 2020 research and innovation programme under grant agreement No 766726

THANK YOU FOR YOUR ATTENTION

QUESTIONS?

COMMENTS?