

(Paris, France) Saclay Magneto-Crystalline anisotropy of Fe, Co and Ni slabs : A benchmark from DFT and Tight-Binding models



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MAGNETIC ANISOTROPY

In magnetic materials :

- Easy-axis magnetization : $E_{syst} = f(\vec{m})$
- Switchable domains \rightarrow Data storage.



- → Nanoscale systems.
- → Smaller ? Thermal fluctuations...
- $\rightarrow\,$ Find system with large Anisotropy.

Why an easy-axis instead of another ?-

 \rightarrow We study the difference of total energy under two different magnetizations :

 $MAE = E_{tot}(m_1) - E_{tot}(m_2)$





- Origin is twofold :
- Shape anisotropy.
- Magneto-crystalline anisotropy (MCA).

ea

TWO ANISOTROPIES

- Shape anisotropy
- Two magnetizations \vec{m}_i, \vec{m}_j , as a magnetic dipole.

$$E_{dip} = \frac{\mu_0}{8\pi} \sum_{i \neq j} \frac{1}{r_{ij}^3} [\vec{m}_i \cdot \vec{m}_j - 3 \frac{(\vec{r}_{ij} \cdot \vec{m}_i)(\vec{r}_{ij} \cdot \vec{m}_j)}{r_{ij}^2}]$$

• Dipole-dipole interaction.

$$E_{dip} = \frac{\mu_0}{8\pi} \sum_{i \neq j} \frac{m_i m_j}{r_{ij}^3} (1 - 3\cos^2(\theta_{ij}))$$

- \rightarrow **in-plane** magnetization.
- Independent of crystalline structure.
- Depends only on quantity of matter and shape.
- At big scales, wins always !

- MCA
- H = Schrödinger + Zeeman + Mass velocity + Spin-orbit → relativistic corrections.
- Quantum effects :

$$H^{SOC} = \sum_{i} \xi_{i,d} \vec{L}_{i} \cdot \vec{S}_{i}$$



- SOC breaks spherical invariance.
- MCA depends on symmetry, nature of atoms.
- Dominating at surfaces and interfaces.

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HOW TO CALCULATE MCA (1/2)

• <u>Brute force method (self-consistent)</u>: $MCA = E_{inplane} - E_{outofplane}$ where the two energies are obtained from SCF calculation including SOC.

In principle « exact » but very time consuming and hard to converge One should use penalization techniques to obtain energy for any direction.

• <u>Force Theorem method</u> : The variation of energy between a SCF calculation without SOC and with SOC is just the band energy variation.

$$MCA^{FT} = \int_{0}^{E_{F}^{1}} E\rho_{1}(E) dE - \int_{0}^{E_{F}^{2}} E\rho_{2}(E) dE$$

$$\Delta E_{tot} = \Delta E_{band}$$

 $MCA^{FTgc} = \int_{0}^{E_{F}} (E - E_{F}) \Delta \rho(E) dE$, at fixed chemical potential.

Very « fast » and numerically stable but cannot be applied to systems with too large SOC.



HOW TO CALCULATE MCA (2/2)

- Three computational tools : home-made Tight-Binding (TB), Quantum Espresso (DFT QE) and Quantum ATK (DFT QATK).
 - $\rightarrow\,$ experimentally, MCA really weak, numerically too ! We have to compare different approaches.

TΒ

Semi-empirical Tight-Binding fitted on Density Functionnal Theory data :

- · On-site
- · Hopping
- · Overlap
- · Stoner parameter
- SOC

 \rightarrow Really fast computational time.



<u>Quantum Espresso</u> : *Ab initio* expanded on plane waves.

→ complete basis describing the whole system plus void, but huge computational cost.

DFT



<u>Quantum ATK</u> : *Ab initio* expanded on Local Atomic Orbitals.

 → localized basis with a really good description of partial system, fast computational time.

TOTAL MCA FOR SLABS : IRON, COBALT AND NICKEL (1/2)

	Fe bcc (001,110,1 11)	Co fcc (001,110,1 11)	Co hcp (0001)	Ni fcc (001,110,1 11)
Lattice parameter (Angström)	a=2,8665	a=3,5447	a=2,5071 c=4,0686	a=3,5249



Mesh SCF : 25*25 k-points / Mesh NSCF : 50*50 k-points

TOTAL MCA FOR SLABS : IRON, NICKEL AND COBALT (2/2)







Co fcc and hcp vs thickness (QE)



Fe bcc vs thickness (QATK)

DFT



Ni fcc vs thickness (QATK)



Co fcc and hcp vs thickness (QATK)





MCA BY LAYERS FOR SLABS : IRON, NICKEL AND COBALT(1/2)

• How to explain oscillations ? It can be interesting to decompose MCA, since we can write :

$$MCA^{FTgc} = \int_{0}^{E_{F}} (E - E_{F}) \Delta \rho(E) dE = \sum_{i,\lambda,k} \int_{0}^{E_{F}} (E - E_{F}) \Delta \rho_{i,\lambda,k}(E) dE$$

- \rightarrow decomposition by layers of MCA, index i
- \rightarrow decomposition by orbitals, index λ
- → large thickness behavior
- \rightarrow MCA in Brillouin zone, index k
- → band structure

Co hcp vs site (TB)



MCA BY LAYERS FOR SLABS : IRON, NICKEL AND COBALT(2/2)







7 8 9 Site (nth layer) 10 11

▲ Ni fcc 111 ● ● Ni fcc 110 ■ ■ Ni fcc 001

12 13 14 15

MAE (meV)

-0 '

-0.8

2

4





Co fcc and hcp d orbital vs site (QATK)



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MCA ANALYSIS IN TIGHT-BINDING : LARGE THICKNESS BEHAVIOR FOR COBALT



MCA ANALYSIS IN TIGHT-BINDING : K-SPACE FOR BULK COBALT





MCA ANALYSIS IN TIGHT-BINDING : BAND STRUCTURE FOR BULK COBALT



Most important contributions : high symmetry points H and Γ !

 \rightarrow tangential bands at Fermi level in one orientation but not in the other one.

• According to the level, MCA varies rapidly !

BAND STRUCTURE IN DFT FOR BULK COBALT





Differences observed in MCA between codes = Slightly differences in band structures according to Fermi level !

FEW IDEAS ABOUT TUNING MCA



Modifying the filling is a way to pilot MCA from in-plane to out-of-plane orientation :

- Electric field \rightarrow 1.10^10 V/m for only 0,05 electrons...
- · Molecules and charge transfer.





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THANK YOU FOR YOUR ATTENTION

QUESTIONS?

COMMENTS?